

SIMPLE MODEL OF LATTICE DYNAMICS FOR MAGNESIUM,
COBALT AND YTRIUM

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A simple model of lattice dynamics for hexagonal, close-packed structure crystals is proposed. Parameters of the model can be derived from the elastic constants data. The model was successfully tested by evaluating phonon frequencies for magnesium, and comparing them with experimental ones. Using the model, phonon dispersion curves and frequency spectra for magnesium, cobalt and yttrium are calculated.

Introduction

Phonon dispersion curves for hexagonal, close-packed (hcp) structure metals Be, Mg and Zn have been measured at room temperatures some years ago [1-5] and many theoretical treatments of these results, both phenomenological and "from first principles" have been published [6-10]. However, the lattice dynamics of many other hcp metals is still unknown, but only in the acoustic $\vec{q} \rightarrow 0$, $\omega \rightarrow 0$ limit (the elastic constants). It seems interesting to try obtaining more information about the phonon spectrum for these metals by a proper extrapolation of the acoustic limit information to other (\vec{q}, ω) regions.

A simple model of the Born-v. Karman type for the lattice dynamics of hexagonal, close-packed structure is presented below. The main advantage of the model is that its parameters can be fixed by using all the elastic constants and the elastic constants only. The model works relatively well for magnesium, and one can expect it to work well also for cobalt and yttrium. The model was partly suggested by the results of Champier and Touissant [11].

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Description of the model

All five elastic constants of the hcp structure can be in general expressed as non-linear combinations of force constants of the Born-v. Karman theory [12]. There exists also an interrelation between force constants, the so-called Huang condition for the absence of nonisotropic internal stresses. Using these six equations one can determine no more than six force constants. They are just six, if the equations are linear and consistent.

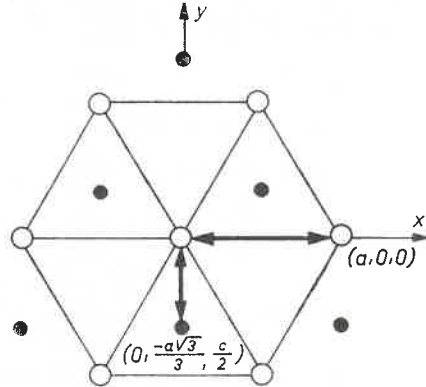


Fig. 1. Projection of the hcp structure on the basal plane and the scheme of interactions for the model discussed

A following model yields a set of equations which underlie these conditions (Fig. 1, Table I):

- i. Interactions of a general type with nearest neighbours in the basal plane $z = 0$.
- ii. "Quasi-central" force matrix for interactions with nearest neighbours in the adjacent, $z = \pm \frac{c}{2}$ planes. This resembles a central forces matrix because the xx -diagonal force constant is assumed equal to 0, but the remaining force constants B, G, D are symmetry-restricted only.

TABLE I

Position of the representative atom of a coordination sphere	Indices in the scheme of paper [12]	Force matrix in the model presented
$\begin{bmatrix} a \\ 0 \\ 0 \\ 0 \\ -\frac{a\sqrt{3}}{3} \\ \frac{c}{2} \end{bmatrix}$	$\begin{aligned} k &= 0 \\ l &= 1 \\ m &= 0 \\ \\ K &= 0 \\ L &= 0 \\ M &= 0 \end{aligned}$	$- \begin{bmatrix} a & f & 0 \\ -f & b & 0 \\ 0 & 0 & g \end{bmatrix}$ $- \begin{bmatrix} 0 & 0 & 0 \\ 0 & B & D \\ 0 & D & G \end{bmatrix}$

The antisymmetric force constant f cannot be fixed from the elastic constants alone, for it does not appear in the corresponding equations. We have left it in the formulas for

elements of dynamical matrix, because it represents interesting many-body forces (f becomes equal to zero when only two-particle ion-ion interaction potentials occur).

The elastic constants and the Huang condition can be written as follows (see [12])

$$c_{11} = \frac{\sqrt{3}}{2c} \cdot \left(3 \cdot a + b + \frac{2}{3} \cdot B \right)$$

$$c_{33} = \frac{\sqrt{3}c}{a^2} \cdot G$$

$$c_{44} = \frac{2}{\sqrt{3}c} \cdot (3 \cdot g + G)$$

$$c_{66} = \frac{\sqrt{3}}{2c} \cdot (a + 3 \cdot b)$$

$$c_{13} + c_{44} = \frac{2}{a} \cdot D$$

$$\frac{3}{4} \cdot \left(\frac{c}{a} \right)^2 \cdot B = 3 \cdot g + G.$$

As a solution one obtains

$$a = \frac{\sqrt{3}c}{4} \cdot c_{11} - \frac{1}{2\sqrt{3}} \cdot \frac{a^2}{c} \cdot c_{44} - \frac{c}{4\sqrt{3}} \cdot c_{66}$$

$$b = \frac{\sqrt{3}c}{4} \cdot c_{66} + \frac{1}{6\sqrt{3}} \cdot \frac{a^2}{c} \cdot c_{44} - \frac{c}{4\sqrt{3}} \cdot c_{11}$$

$$g = \frac{c}{2\sqrt{3}} \cdot \left(c_{44} - \frac{2}{3} \cdot \left(\frac{a}{c} \right)^2 \cdot c_{33} \right)$$

$$B = \frac{2a^2}{\sqrt{3}c} \cdot c_{44}$$

$$G = \frac{a^2}{\sqrt{3}c} \cdot c_{33}$$

$$D = \frac{a}{2} \cdot (c_{13} + c_{44}).$$

The elements of the dynamical matrix are given in the Appendix and phonon dispersion formulae can be found in [21]. In this simple model the following equation holds (group theory classification of phonon modes is according to Iyengar *et al.* [2] and Pynn and Squires [3]):

$$\frac{\omega(\Gamma_3^+)}{\omega(\Gamma_5^+)} = \sqrt{\frac{c_{33}}{c_{44}}}.$$

If purely central interactions ($g = 0$) and ideal hcp structure $\left(\frac{c}{a} = \sqrt{\frac{8}{3}} \right)$ are assumed, the Huang condition predicts:

$$\frac{c_{33}}{c_{44}} = 4.$$

Phonon dispersion curves

The model was tested by calculating force constants and phonon frequencies for Be, Mg and Zn, and comparing them with experimental data. A satisfactory fit was obtained only for magnesium (Figs 2, 3). Optical frequencies at $\vec{q} = 0$ fit experimental values very well: $\omega(\Gamma_3^+) = 4.59_{10}^{13}$ rad/sec, $\omega(\Gamma_5^+) = 2.37_{10}^{13}$ rad/sec.

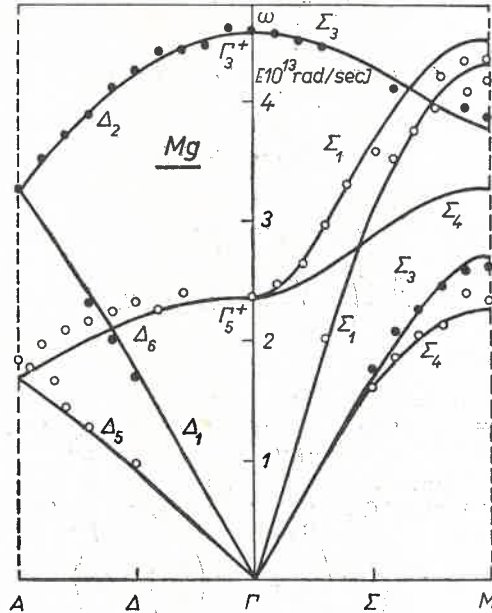


Fig. 2. Phonon dispersion curves in the directions Δ and Σ for Mg. Points — experimental data of Iyengar *et al.* [2]. Lines — model discussed

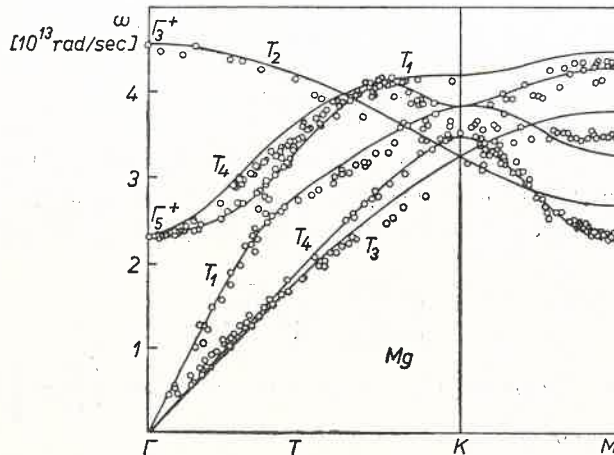


Fig. 3. Phonon dispersion curves in the direction T for Mg. Points — experimental data of Squires and Pynn [3], at Γ and M — data of Iyengar *et al.* [2]. Lines — model discussed

In fact, for Mg the present model fits the experiment nearly as well as the Iyengar's axial symmetry model and the four-nearest-neighbours general force model, where some force constants were found by least-square fitting to experimental phonon frequencies [2], and the three-nearest-neighbours general force model of Squires [3]. For the direction T the agreement of our model with experiment is somewhat worse, although the discrepancies do not exceed $\sim 5\%$. For branches T_2 and T_4 acoustic the agreement is still good.

There is also another, more sophisticated model of lattice dynamics of Mg, based on the elastic constants only [7]. The interactions with 4 nearest neighbours and the contribution of electronic bulk modulus to lattice dynamics are taken into account. In spite of greater complexity, the agreement of calculated dispersion curves with experiment is not better than in our case and for some phonon frequencies (ex. $\omega(\Gamma_3^+)$) — slightly worse.

The ratios of elastic constants, c_{44}/c_{66} and c_{33}/c_{11} for Mg are nearly equal to one, which is characteristic for isotropic media. Also the ratio c/a for magnesium (1.62) is close to the c/a ratio for ideal hcp structure (1.633). As may be seen (Table II), the same is true also

TABLE II
Lattice constants (Å) and elastic constants (10^{11} dyn/cm²)

	Mg	Co	Y
a	3.2028	2.51	3.65
c	5.196	4.07	5.73
c_{11}	5.964	30.07	7.79
c_{33}	6.164	35.81	7.69
c_{66}	1.702	7.1	2.47
c_{44}	1.642	7.55	2.43
c_{13}	2.104	10.27	5.51

(Elastic constants for Mg and Co from [14], for Y from [15]).

TABLE III
Force constants (dyn/cm) for Mg, Co and Y

a	b	g	B	G	D
Mg $1.12_{10}4$	$-3.31_{10}2$	$1.21_{10}2$	$3.74_{10}3$	$7.03_{10}3$	$6.06_{10}3$
Co $4.55_{10}4$	$-4.03_{10}3$	$-1.80_{10}3$	$1.35_{10}4$	$3.20_{10}4$	$2.24_{10}4$
Y $1.57_{10}4$	$2.30_{10}2$	$5.79_{10}2$	$6.52_{10}3$	$1.03_{10}4$	$8.27_{10}3$

for cobalt and yttrium. Assuming that these macroscopic similarities reflect some analogies in their microscopic interactions, one can expect similarities in other properties too. Based on this assumption, the force constants and phonon dispersion curves for Co and Y were calculated (Table III, Figs 4, 5). The optical frequencies at $\vec{q} = 0$, $\omega(\Gamma_3^+)$, $\omega(\Gamma_5^+)$ are as follows (the unit- rad/sec):

$$\text{Co: } 6.23_{10}13, 2.86_{10}13; \text{ Y: } 2.61_{10}13, 1.47_{10}13.$$

At present there are no experimental data to be compared with these values.

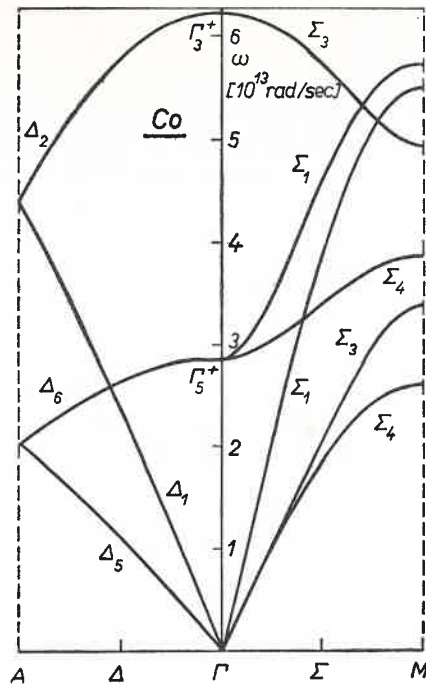


Fig. 4. Model calculated phonon dispersion curves in the directions Δ and Σ for Co

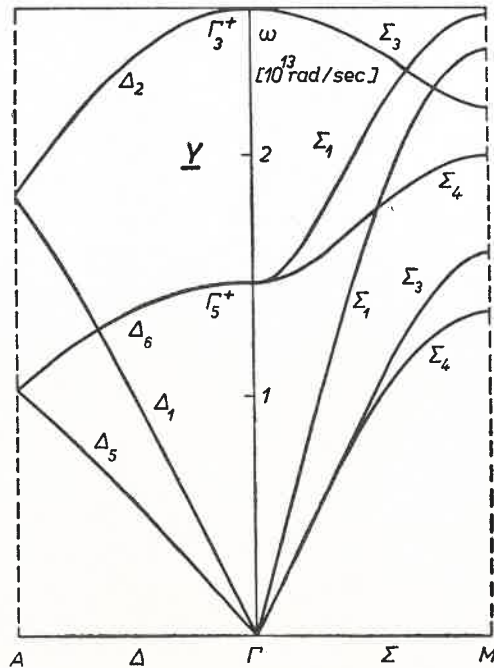


Fig. 5. Model calculated phonon dispersion curves in the directions Δ and Σ for Y

Determination of the phonon frequency spectra

Having developed the model, one is in a position to calculate phonon frequencies and polarization vectors throughout all the Brillouin zone. This information is needed for evaluating many quantities of physical interest. For example, even with a dynamical matrix based on a rather rough model, one obtains polarization vectors and phonon frequencies sufficiently accurately to estimate the dynamical form-factor, necessary for planning the neutron-phonon scattering experiments [2]. Other quantities of this kind are the probability of the Mössbauer effect, particularly in the case of anisotropy [17], or the electron diffuse scattering intensity [16]. The knowledge of the frequency spectrum permits for the calculation of thermodynamic properties of a crystal to be performed [18].

To calculate the frequency spectrum we used the sampling method improved by Gilat and Dolling [19]. In our case the irreducible part of I BZ was divided into a mesh of points, the number of which was about 80. At these points the exact eigenfrequencies and eigenvectors of the dynamical matrix (see Appendix) were computed. Following this, the perturba-

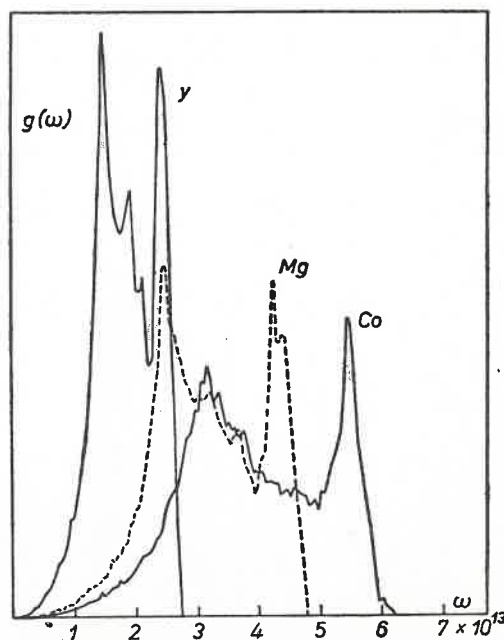


Fig. 6. The averaged phonon frequency distributions for Mg, Co and Y. $\Delta\omega = 5_{10}11$ rad/sec

tion calculations of the frequencies at 125 points, uniformly distributed within a cube about every mesh point, were carried out. Frequencies obtained in this way, usually about 30000 in number were sorted into frequency intervals $\Delta\omega = 5_{10}11$ rad/sec or $\Delta\omega = 1_{10}12$ rad/sec. Every frequency was given a weight (1 or less) following from symmetry considerations. After summing up all the weights in every frequency interval the phonon frequency distribution function (FD) was established.

To avoid, partly at least, the irrelevant peaks in FD due to accidental degeneracies or weight inaccuracies, we repeated all the computations with the mesh of points shifted along the diagonal of a cube by a half of a diagonal, and to add the newly obtained FD-II to the former one, FD-I. The so-obtained "averaged" FD was always smoother than any of the original ones — many small peaks were usually removed. The GIER computer at INR-Świerk was used to perform all the computations.

The averaged frequency distribution for Mg, computed on the assumption $f = 0$ is shown in Fig. 6. The quality of this result is lower than that obtained by Raubenheimer and Gilat [20], as the mesh in their work was much denser than ours, resulting in better accuracy and resolution.

Besides the quality, both slopes differ also in their physical content. In our case the peak at $\omega = 2.5_{10}^{13}$ rad/sec is the highest, whereas they obtained in this region only a maximum, much lower and broader, with some structure. This is, of course, due to the differences

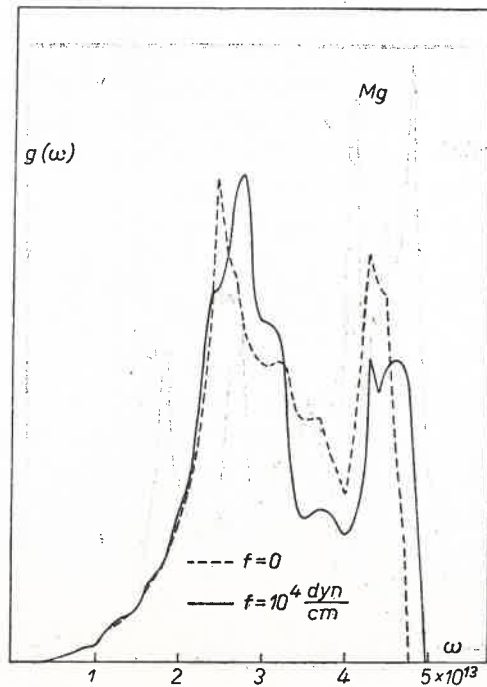


Fig. 7. The phonon frequency distribution FD-I, for Mg, assuming $f = 0$ and $f = 10^4$ dyn/cm, $\Delta\omega = 10^{12}$ rad/sec

in the models used. Raubenheimer and Gilat rely on the four neighbours, tensor-force model of Iyengar *et al.* [2], which at first sight seems better in every respect than ours, because it was fitted to a larger number of experimental data. However, it predicts a near degeneracy of two frequencies at point *M* (see [20]), which disagrees with experimental phonon data. There is no such effect in our model. Also at point *K* the present model clearly fits the experimental data better. For these reasons we found it interesting to determine

the frequency distribution for Mg also within the present model. However, to obtain a fully reliable FD, one should repeat the computations with still better models, starting perhaps with the one given by Squires [19].

Fig. 6 shows the averaged FD's also for Co and Y. It is interesting to note that the distributions cannot be obtained one from another by any ω -scaling. In the case of Co the total number of frequencies to be sorted was relatively lower, for the $(0, \omega_{\max})$ range is in this case the longest. That is why in the case of Co even the "averaging" procedure did not remove many of the obviously accidental peaks.

The tables of polarization vectors and phonon frequencies corresponding to the mesh points are given in [22].

Antisymmetric, many-body force constant cannot be determined from the phonon data for symmetry directions. It seems that this quantity has never been studied systematically. As it does not appear in the expressions for elastic constants, it cannot influence the low-frequency part of FD, and therefore the low-temperature thermodynamic properties of crystals.

The unaveraged, FD-I slopes for Mg are shown in Fig. 7. Frequency distributions for $f = 10^3$ dyn/cm and $f = 0$ are practically indistinguishable. The $f = 10^4$ dyn/cm FD is already modified a much, excepting at the low frequency range. The main peaks became broader and shifted to the right. The ω_{\max} increased considerably, as compared with the $f = 0$ case. This means that highest frequencies occur in this case at the points of the I BZ, which do not lie along the main symmetry directions.

Discussion

The relative success of the model presented is obviously due to the domination of the short range ion-ion interactions in the lattice dynamics of Mg. We notice (Table III) that $\sum_{\alpha} \Phi_{\alpha\alpha}(1, 1) \simeq \sum_{\alpha} \Phi_{\alpha\alpha}(1, 2)$, i.e. this interaction is highly isotropic.

In terms of isotropic, two-body interaction potential $V(r)$, the smallness of the force constants b and g $\left(\sim \left(\frac{1}{r} \frac{\partial V}{\partial r} \right)_{r=a} \right)$ suggests, that corresponding ions occupy positions very close to the minimum of $V(r)$. This is consistent with the short-range interaction concept. It follows from our estimations, that a simple potential

$$V(r) = \begin{cases} -\delta + \frac{\alpha}{2} (r-r_0)^2 & \text{at the two nearest atoms positions,} \\ 0 & \text{besides,} \end{cases}$$

(where δ , α , and r_0 are adjustable parameters), enables one to reproduce the experimental phonon curves for Mg with the accuracy $\sim 8\%$. By allowing a small anisotropy in the potential, one could probably obtain an even better agreement. The parameter δ can be determined from cohesion energy.

It follows from the present work that for Mg, and perhaps for Co and Y, most of the information concerning the form of the microscopic interactions can be deduced from

TABLE IV

Averaged phonon frequency spectrum for Mg

$$\Delta\omega = 5_{10}11 \text{ rad/sec, } \sum_i g(\omega_i) = 1$$

0.000	1.149 ₁₀ -3	8.748 ₁₀ -3	1.776 ₁₀ -2	1.052 ₁₀ -2
0.000	1.474 ₁₀ -3	8.935 ₁₀ -3	1.801 ₁₀ -2	1.259 ₁₀ -2
5.380 ₁₀ -6	1.774 ₁₀ -3	9.905 ₁₀ -3	1.747 ₁₀ -2	1.355 ₁₀ -2
6.311 ₁₀ -6	1.873 ₁₀ -3	1.137 ₁₀ -2	1.774 ₁₀ -2	1.605 ₁₀ -2
9.560 ₁₀ -6	2.089 ₁₀ -3	1.295 ₁₀ -2	1.823 ₁₀ -2	2.198 ₁₀ -2
5.753 ₁₀ -5	2.519 ₁₀ -3	1.421 ₁₀ -2	1.780 ₁₀ -2	2.724 ₁₀ -2
5.808 ₁₀ -5	2.348 ₁₀ -3	1.706 ₁₀ -2	1.665 ₁₀ -2	2.236 ₁₀ -2
9.467 ₁₀ -5	2.708 ₁₀ -3	1.909 ₁₀ -2	1.590 ₁₀ -2	2.224 ₁₀ -2
1.374 ₁₀ -4	2.736 ₁₀ -3	2.118 ₁₀ -2	1.540 ₁₀ -2	2.290 ₁₀ -2
1.770 ₁₀ -4	3.231 ₁₀ -3	2.807 ₁₀ -2	1.472 ₁₀ -2	2.322 ₁₀ -2
2.576 ₁₀ -4	3.299 ₁₀ -3	2.843 ₁₀ -2	1.453 ₁₀ -2	1.932 ₁₀ -2
2.693 ₁₀ -4	3.693 ₁₀ -3	2.461 ₁₀ -2	1.387 ₁₀ -2	1.426 ₁₀ -2
3.849 ₁₀ -4	4.733 ₁₀ -3	2.303 ₁₀ -2	1.438 ₁₀ -2	1.120 ₁₀ -2
3.875 ₁₀ -4	4.560 ₁₀ -3	2.230 ₁₀ -2	1.428 ₁₀ -2	8.375 ₁₀ -3
7.616 ₁₀ -4	5.099 ₁₀ -3	2.111 ₁₀ -2	1.500 ₁₀ -2	6.382 ₁₀ -3
6.964 ₁₀ -4	5.078 ₁₀ -3	2.056 ₁₀ -2	1.270 ₁₀ -2	2.124 ₁₀ -3
6.867 ₁₀ -4	5.883 ₁₀ -3	1.916 ₁₀ -2	1.204 ₁₀ -2	
9.918 ₁₀ -4	5.912 ₁₀ -3	1.861 ₁₀ -2	1.145 ₁₀ -2	
1.180 ₁₀ -3	7.416 ₁₀ -3	1.825 ₁₀ -2	1.080 ₁₀ -2	
1.080 ₁₀ -3	7.847 ₁₀ -3	1.733 ₁₀ -2	9.856 ₁₀ -3	

TABLE V

Averaged phonon frequency spectrum for Co

$$\Delta\omega = 5_{10}11 \text{ rad/sec, } \sum_i g(\omega_i) = 1$$

0.000	7.429 ₁₀ -4	3.634 ₁₀ -3	1.669 ₁₀ -2	1.174 ₁₀ -2	8.441 ₁₀ -3	1.198 ₁₀ -3
0.000	7.667 ₁₀ -4	4.132 ₁₀ -3	1.641 ₁₀ -2	1.145 ₁₀ -2	1.007 ₁₀ -2	1.460 ₁₀ -3
1.342 ₁₀ -6	9.245 ₁₀ -4	4.678 ₁₀ -3	1.908 ₁₀ -2	1.163 ₁₀ -2	9.866 ₁₀ -3	9.189 ₁₀ -4
1.825 ₁₀ -5	9.109 ₁₀ -4	5.014 ₁₀ -3	1.879 ₁₀ -2	1.071 ₁₀ -2	1.084 ₁₀ -2	6.668 ₁₀ -4
1.552 ₁₀ -5	1.056 ₁₀ -3	4.997 ₁₀ -3	2.045 ₁₀ -2	1.094 ₁₀ -2	1.179 ₁₀ -2	4.195 ₁₀ -4
3.500 ₁₀ -5	1.278 ₁₀ -3	5.058 ₁₀ -3	1.964 ₁₀ -2	1.087 ₁₀ -2	1.256 ₁₀ -2	2.022 ₁₀ -5
4.841 ₁₀ -5	1.296 ₁₀ -3	5.702 ₁₀ -3	1.734 ₁₀ -2	1.059 ₁₀ -2	1.336 ₁₀ -2	
2.286 ₁₀ -5	1.425 ₁₀ -3	6.287 ₁₀ -3	1.761 ₁₀ -2	1.078 ₁₀ -2	1.406 ₁₀ -2	
6.749 ₁₀ -5	1.333 ₁₀ -3	6.531 ₁₀ -3	1.828 ₁₀ -2	9.857 ₁₀ -3	1.568 ₁₀ -2	
3.918 ₁₀ -5	2.046 ₁₀ -3	7.135 ₁₀ -3	1.649 ₁₀ -2	1.008 ₁₀ -2	1.914 ₁₀ -2	
8.341 ₁₀ -5	1.661 ₁₀ -3	8.218 ₁₀ -3	1.532 ₁₀ -2	1.028 ₁₀ -2	2.414 ₁₀ -2	
1.944 ₁₀ -4	1.849 ₁₀ -3	9.305 ₁₀ -3	1.571 ₁₀ -2	9.599 ₁₀ -3	2.359 ₁₀ -2	
2.293 ₁₀ -4	2.222 ₁₀ -3	9.742 ₁₀ -3	1.502 ₁₀ -2	1.046 ₁₀ -2	1.905 ₁₀ -2	
2.078 ₁₀ -4	2.468 ₁₀ -3	1.025 ₁₀ -2	1.531 ₁₀ -2	1.022 ₁₀ -2	1.466 ₁₀ -2	
3.496 ₁₀ -4	2.420 ₁₀ -3	1.038 ₁₀ -2	1.451 ₁₀ -2	9.951 ₁₀ -3	1.254 ₁₀ -2	
3.575 ₁₀ -4	2.939 ₁₀ -3	1.169 ₁₀ -2	1.438 ₁₀ -2	9.745 ₁₀ -3	1.009 ₁₀ -2	
4.599 ₁₀ -4	2.728 ₁₀ -3	1.269 ₁₀ -2	1.505 ₁₀ -2	9.714 ₁₀ -3	8.775 ₁₀ -3	
4.154 ₁₀ -4	2.838 ₁₀ -3	1.427 ₁₀ -2	1.392 ₁₀ -2	8.860 ₁₀ -3	7.465 ₁₀ -3	
5.860 ₁₀ -4	3.133 ₁₀ -3	1.454 ₁₀ -2	1.244 ₁₀ -2	8.971 ₁₀ -3	5.436 ₁₀ -3	
6.336 ₁₀ -4	3.395 ₁₀ -3	1.526 ₁₀ -2	1.145 ₁₀ -2	9.492 ₁₀ -3	2.531 ₁₀ -3	

TABLE VI

Averaged phonon frequency spectrum for Y

$$\Delta\omega = 5_{10}11 \text{ rad/sec, } \sum_i g(\omega_i) = 1$$

0.000	8.080 ₁₀ -3,	3.033 ₁₀ -2,
1.493 ₁₀ -5,	9.249 ₁₀ -3,	2.643 ₁₀ -2,
4.849 ₁₀ -5,	1.098 ₁₀ -2,	2.642 ₁₀ -2,
7.935 ₁₀ -5,	1.304 ₁₀ -2,	2.774 ₁₀ -2,
1.314 ₁₀ -4,	1.431 ₁₀ -2,	2.377 ₁₀ -2,
3.560 ₁₀ -4,	1.757 ₁₀ -2,	2.038 ₁₀ -2,
4.504 ₁₀ -4,	2.109 ₁₀ -2,	2.125 ₁₀ -2,
7.935 ₁₀ -4,	2.551 ₁₀ -2,	2.803 ₁₀ -2,
8.087 ₁₀ -4,	3.118 ₁₀ -2,	3.914 ₁₀ -2,
1.312 ₁₀ -3,	3.863 ₁₀ -2,	4.473 ₁₀ -2,
1.645 ₁₀ -3,	4.765 ₁₀ -2,	4.214 ₁₀ -2,
2.077 ₁₀ -3,	4.364 ₁₀ -2,	3.770 ₁₀ -2,
2.311 ₁₀ -3,	3.869 ₁₀ -2,	2.229 ₁₀ -2,
2.914 ₁₀ -3,	3.503 ₁₀ -2,	1.276 ₁₀ -2,
3.388 ₁₀ -3,	3.299 ₁₀ -2,	1.800 ₁₀ -3,
4.342 ₁₀ -3,	3.155 ₁₀ -2,	
4.664 ₁₀ -3,	3.112 ₁₀ -2,	
5.332 ₁₀ -3,	3.240 ₁₀ -2,	
6.173 ₁₀ -3,	3.240 ₁₀ -2,	
7.921 ₁₀ -3,	3.487 ₁₀ -2,	

the elastic constants. It seems possible, that the "first principles" theories of lattice dynamics for the hcp metals [9, 10] could be successfully adapted to the case of Co and Y, by varying their adjustable parameters to fit the theory and experiment at the acoustic limit only.

Note added in proof: The presented phonon dispersion curves for yttrium show a qualitative agreement with the recently published experimental phonon data of S. K. Sinha *et al.* (*Phys. Rev. B1*, 2430 (1970)).

APPENDIX

Elements of dynamical matrix

$$md_{11}(\vec{q}) = 3 \cdot B + 3 \cdot (a+b) - 2 \cdot a \cdot \cos(2k_x) - (a+3b) \cdot \cos(k_x) \cdot \cos(k_y)$$

$$md_{22}(\vec{q}) = 3 \cdot B + 3(a+b) - 2b \cdot \cos(2k_x) - (3a+b) \cdot \cos(k_x) \cdot \cos(k_y)$$

$$md_{33}(\vec{q}) = 6 \cdot G - 2 \cdot g \cdot (3 - \cos(2k_x) - 2 \cdot \cos(k_x) \cdot \cos(k_y))$$

$$md_{12}(\vec{q}) = \sqrt{3} \cdot (a-b) \cdot \sin(k_x) \cdot \sin(k_y) + i \cdot 2f \cdot (\sin(2k_x) - 2 \cdot \sin(k_x) \cdot \cos(k_y))$$

$$md_{13}(\vec{q}) = md_{23}(\vec{q}) = 0$$

$$mD_{11}(\vec{q}) = -3 \cdot B \cos(k_x) \cdot \cos(k_x) \cdot e^{i(k_z - k_y)}$$

$$mD_{22}(\vec{q}) = -B \cdot \cos(k_x) \cdot (2 \cdot e^{ik_z} + \cos(k_x) \cdot e^{i(k_z - k_y)})$$

$$mD_{33}(\vec{q}) = -2 \cdot G \cdot \cos(k_x) \cdot (e^{ik_z} + 2 \cdot \cos(k_x) \cdot e^{i(k_z - k_y)})$$

$$mD_{12}(\vec{q}) = i \cdot \sqrt{3} \cdot B \cdot \cos(k_z) \cdot \sin(k_x) \cdot e^{i(k_z - k_y)}$$

$$mD_{13}(\vec{q}) = -2 \cdot \sqrt{3} \cdot D \cdot \sin(k_z) \cdot \sin(k_x) \cdot e^{i(k_z - k_y)}$$

$$mD_{23}(\vec{q}) = i \cdot 2 \cdot D \cdot \sin(k_z) \cdot (e^{ik_z} - \cos(k_x)) \cdot e^{i(k_z - k_y)}$$

$$k_x = \pi a q_x \quad k_y = \sqrt{3} \pi a q_y \quad k_z = \pi c q_z$$

$$D_{\alpha\beta}(\vec{q}|11) = D_{\beta\alpha}^*(\vec{q}|11) = D_{\alpha\beta}^*(\vec{q}|22) = d_{\alpha\beta}(\vec{q})$$

$$D_{\alpha\beta}(\vec{q}|12) = D_{\beta\alpha}(\vec{q}|12) = D_{\alpha\beta}^*(\vec{q}|21) = D_{\alpha\beta}(\vec{q}).$$

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